

applied force and the net plane used were the same for Fig. 3 as for Fig. 2.

Position *a* covers the regions *F* and *D*. Elliptic and hook-shaped fringes are recognized, although a parallel-sided crystal was used. Similar ones appear also in the top and bottom parts of sections *b* and *c*. Incidentally, if the crystal is in the shape of a parallel slab and undistorted, a system of parallel fringes should be expected (Kato, 1961*a*, *b*; Authier & Lang, unpublished). The appearance of non-parallel fringes is understandable by the dynamical diffraction theory on slightly deformed crystals (Kato, 1963, 1964*a*, *b*). According to the theory, the fringe distance should decrease with increase of the strain gradient.* In fact, the fringe position is determined by an effective thickness *Z* which is proportional to the real thickness and the strain gradient (*cf.* equation (30) of Kato, 1964*b*). Since the strain gradient is maximum on the central line *PP'* of Fig. 2 in every column section of the crystal, higher order fringes will spring up from the central part of the section pattern. In other words, the central part is effectively thick. Since the fringes in the traverse pattern are essentially the loci of the apices of the hook-shaped and elliptic fringes, they may be called the 'fringes of equal strain-gradient', just as the terms 'equal thickness fringes' and 'equal inclination fringes' are used in electron microscopy. The fact that the behaviour of fringe phenomena is the same for the *hkl* and *hkl* reflexions is also expected from the theory.

The dark background in the section pattern *a* accounts for the appearance of the region *D* in the traverse pattern. In general, the darkness of traverse patterns corresponds to the integrated intensity of crystals. The enhancement of the integrated intensity in distorted crystals is expected from the previous theory (Kato, 1964*b*). According to the theory, however, the departure from Friedel's law can not be

* Although the conclusion was derived in the special case of homogeneous bending, it can be shown that the theory is applicable correctly to any type of small distortion in which the displacement can be represented in a quadratic form of coordinates (Kato & Ando, 1966).

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Bond lengths and thermal vibrations in *m*-dinitrobenzene. BY JAMES TROTTER* and C.S. WILLISTON, *Department of Chemistry, University of British Columbia, Vancouver 8, B.C., Canada*

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X-ray analysis of *m*-dinitrobenzene crystals (Trotter, 1961*a*, *b*) has established the main features of the molecular structure: the carbon and nitrogen atoms all lie in one plane, but the nitro groups are twisted about 11° out of this plane. The bond distances and valency angles were, however, less precisely measured, and showed variations which were unlikely to be real. A full-matrix least-squares refinement (unpublished), in which a scale-factor parameter for each layer was refined, suggested that much of the trouble in obtaining satisfactory results might be associated with inaccuracies in the scale factors relating the various *hkL* layers (*L* = 0 → 3), rather than in the refinement methods, and to overcome possible errors a new set of intensity data has now been measured by counter methods, and more accurate values

expected unless the Borrmann absorption is taken into account. In the case of our region *D* the Borrmann absorption has nothing to do with the observation, because the sign of the indices for the net plane in which the enhancement is observed is opposite to the sign predicted by the theory under the experimental conditions described above.

In the central part of section pattern *b*, a broad line of high intensity is recognized. This broad line appears also in section pattern *c*. From the geometrical correspondence between the section and traverse patterns we know that the region *G* corresponds to the central broad line in the section pattern. For this reason the regions *D* and *G* were distinguished above, although they look similar in the traverse pattern so far as the intensity is concerned.

Finally, in section pattern *c*, a pair of dark images appears on either side of the central dark line. In pattern *b* the corresponding one is recognized markedly only on the right side of the central line. These pair images correspond to the region *E* in the traverse pattern.

In conclusion, the fringe phenomena can be explained by assuming the contraction of fringe distance in distorted crystals. The contraction has been predicted theoretically. The regions *D*, *E* and *G* can not be understood straightforwardly by the previous theory.

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of the bond distances and angles and of the thermal parameters have been obtained.

The cell parameters, remeasured on the General Electric Goniostat, are: *a* = 13.257, *b* = 14.048, *c* = 3.806 Å. The intensities of all 823 reflexions with $2\theta(\text{Cu } K\alpha) \leq 146^\circ$ (minimum *d* = 0.81 Å) were measured with a scintillation counter, using the θ - 2θ scan; 783 reflexions had intensities above background. The structure was refined by block-diagonal least-squares, minimizing $\sum w(|F_o| - |F_c|)^2$, with $\sqrt{w} = |F_o|/3$ when $|F_o| < 3$ and $\sqrt{w} = 1$ when $|F_o| \geq 3$. Hydrogen atoms were refined with isotropic, and the heavier atoms with anisotropic thermal parameters; the final *R* was 0.097 for the observed reflexions (structure factors are listed in Table 1, and comparison with the previous *F*_o values (Trotter, 1961*a*) indicates that the inter-layer scaling in the previous data contained inaccuracies). The final positional and thermal

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Table 2. Final fractional positional parameters w.r. to crystal axes ($\times 10^4$ for C, N, O; $\times 10^3$ for H), and thermal parameters, U_{ij} being the components of the mean square vibration tensors (in 10^{-2} \AA^2) w.r. to the molecular axes 1, 2 and 3 (Fig. 1)

Atom	<i>x</i>	<i>y</i>	<i>z</i>	U_{11}	U_{12}	U_{13}	U_{22}	U_{23}	U_{33}
C(1)	1399	3652	-0298	3.99	2.37	-0.43	4.31	1.32	5.12
C(2)	1800	4493	-1473	4.30	1.83	0.82	4.08	-1.32	5.00
C(3)	2828	4576	-1198	4.55	1.86	-0.60	4.77	1.19	4.85
C(4)	3450	3860	-0037	6.33	4.26	1.17	5.92	-0.19	7.89
C(5)	3000	3042	1257	3.48	3.46	0.03	9.21	2.35	8.46
C(6)	1971	2934	1101	5.97	1.39	2.10	5.35	-1.50	7.63
N(1)	0288	3531	-0416	7.64	3.02	0.78	4.81	-0.50	7.64
N(2)	3297	5469	-2506	8.72	1.24	1.29	5.76	-1.09	7.42
O(1)	-0067	2850	1268	11.48	4.07	-1.01	4.60	-0.39	17.33
O(2)	-0200	4108	-1986	6.97	5.51	-0.70	7.59	0.31	11.72
O(3)	2756	6009	-4133	9.91	-0.26	0.57	6.74	-1.58	11.58
O(4)	4161	5627	-1512	13.72	6.35	2.70	4.44	-0.40	17.81
H(2)	139	489	-360	$B = 5.0 \text{ \AA}^2$					
H(4)	409	399	029	4.3					
H(5)	326	246	080	8.9					
H(6)	164	239	141	9.6					

Standard deviations

C, N, O	$\sigma(x) = 0.0057\text{--}0.0075 \text{ \AA}$	H	$\sigma(x) = 0.07\text{--}0.09 \text{ \AA}$
	$\sigma(y) = 0.0056\text{--}0.0080$		$\sigma(y) = 0.07\text{--}0.09$
	$\sigma(z) = 0.0082\text{--}0.0127$		$\sigma(z) = 0.09\text{--}0.13$
	$\sigma(U_{ij}) = 0.003\text{--}0.007 \text{ \AA}^2$		$\sigma(B) = 2.0\text{--}3.2 \text{ \AA}^2$

Table 3. Analysis of thermal motion

Aromatic ring treated as a rigid body

$$T = \begin{pmatrix} 3.82 & 2.53 & 0.24 \\ & 4.15 & 0.16 \\ & & 4.14 \end{pmatrix} 10^{-2} \text{ \AA}^2 \quad \omega = \begin{pmatrix} 27.3 & 1.3 & -9.6 \\ & 34.7 & -3.5 \\ & & 32.5 \end{pmatrix} \text{ deg}^2$$

R.M.S. translations = 0.20, 0.20, 0.20 \AA

R.M.S. oscillations = 5.2°, 5.9°, 5.7°

Values of $\overline{u^2}$ in various directions (10^{-2} \AA^2)

	\perp to molecular plane		\parallel to N-O bonds		\perp to N-O bonds		\parallel to C-N bonds
			O(1), 6.3 O(4), 6.0	O(2), 7.6 O(3), 8.7	6.5 7.9	4.8 5.8	4.8 5.6
N(1)	7.6						4.8
N(2)	7.4						5.6
O(1)	17.3		6.5		9.7		—
O(2)	11.7		7.0		7.6		—
O(3)	11.6		9.9		6.7		—
O(4)	17.8		5.4		13.5		—

Table 4. Bond distances and valency angles ($\sigma \sim 0.6^\circ$)

Atoms	Distance (\AA)			Atoms	Angle
	Uncorr.	Corr.	σ		
C(1)–C(2)	1.372	1.384	0.009	C(6)–C(1)–C(2)	123.1°
C(2)–C(3)	1.372	1.384	0.008	C(1)–C(2)–C(3)	115.7
C(3)–C(4)	1.374	1.386	0.009	C(2)–C(3)–C(4)	124.0
C(6)–C(1)	1.369	1.381	0.010	C(3)–C(4)–C(5)	117.6
C(4)–C(5)	1.385	1.397	0.010	C(4)–C(5)–C(6)	120.2
C(5)–C(6)	1.374	1.386	0.009	C(5)–C(6)–C(1)	119.1
C(1)–N(1)	1.484	1.491	0.008	C(2)–C(1)–N(1)	118.3
C(3)–N(2)	1.487	1.494	0.009	C(6)–C(1)–N(1)	118.6
N(1)–O(1)	1.243	1.276	0.009	C(2)–C(3)–N(2)	117.5
N(1)–O(2)	1.197	1.220	0.009	C(4)–C(3)–N(2)	118.4
N(2)–O(3)	1.214	1.230	0.009	C(1)–N(1)–O(1)	116.6
N(2)–O(4)	1.227	1.266	0.009	C(1)–N(1)–O(2)	118.3
				C(3)–N(2)–O(3)	116.8
C(2)–H(2)	1.12		0.07	C(3)–N(2)–O(4)	116.1
C(4)–H(4)	0.87		0.07	O(1)–N(1)–O(2)	125.1
C(5)–H(5)	0.90		0.09	O(3)–N(1)–O(4)	126.5
C(6)–H(6)	0.90		0.09		

and valency angles also conform to symmetry m , and the mean distances and angles are shown in Fig. 1.

The C-C bond distances are all equal within experimental error, the average value of 1.386 ± 0.004 Å not being significantly different from the distance in benzene (1.392 Å). The mean C-N length of 1.493 ± 0.006 Å is similar to that in nitrobenzene, 1.486 ± 0.02 Å (Trotter, 1959). The N-O bond distances average 1.271 ± 0.006 Å for N(1)-O(1) and N(2)-O(4), and 1.225 ± 0.006 Å for N(1)-O(2) and N(2)-O(3). Since these averages differ by 5σ , the structure contains two types of N-O bond, one longer and the other shorter than, with their mean (1.248 Å) equal to, the N-O distance in, for example *p*-nitroaniline, 1.247 ± 0.005 Å (Trueblood, Goldish & Donohue, 1961). The valency angles are all quite similar to those in other nitro compounds. The hydrogen

atom parameters are not very accurate; the mean C-H distance is 0.95 Å.

All the calculations were performed with our own programs on the IBM 7040 computer, and we are indebted to the staff of the U.B.C. Computing Centre for assistance, and the National Research Council of Canada for financial support and for the award of a research studentship to C.S.W.

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Structure of P, P', P'', P''' -Tetrakis(tricarbonylnickel-tetraphosphorus hexaoxide), $P_4O_6[Ni(CO)_3]_4$. By E.D. PIERRON, *Monsanto Physical Sciences Center, St. Louis, Mo., U.S.A.*, P.J. WHEATLEY, *Monsanto Research S.A., Zürich, Switzerland* and J.G. RIESS, *University of Strasbourg, France*

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The reaction of P_4O_6 with an excess of nickel carbonyl produces $P_4O_6[Ni(CO)_3]_4$, a white powder, with evolution of four equivalents of CO. As described elsewhere (Riess & Van Wazer, 1965) the symmetric bird-cage structure of P_4O_6 , which has four unshared pairs of electrons (one on each phosphorus atom) coordinates four nickel tricarbonyl groups.

Since satisfactory single crystals of this substance could not be grown, an attempt has been made to determine the structure from powder intensity data. A General Electric XRD-5 diffractometer with Cu $K\alpha$ radiation was used. Of the 27 measurable diffraction peaks observed up to a maximum $\sin \theta$ value of 0.62, only 19 were due to the diffracted spectrum from a single set of planes.

The pattern could be indexed on the basis of a cubic unit cell with $a = 8.811 \pm 0.003$ Å. The absence of systematic extinctions indicated four possible space groups: $P23$, $Pm3$, $P43m$, and $Pm3m$. The density, determined on the powdered sample with a Beckman air comparison pycnometer, is 1.865 g.cm^{-3} . This value may be compared with a calculated density of 1.919 g.cm^{-3} for one molecule in each unit cell. Accordingly, two of the possible space groups can be eliminated if it is assumed that no disorder is present, since $Pm3$ and $Pm3m$ do not have four equivalent positions.

For the initial trials it was assumed that the correct space group was $P43m$ with the atoms placed in the Wyckoff positions indicated in Table 2. This still leaves two possible molecular configurations which differ merely in a rotation by 60° of the $Ni(CO)_3$ groups. Standard bond lengths and angles (*Tables of Interatomic Distances*, 1958) were taken to calculate a set of coordinates. The only doubtful distance is the length of the Ni---P bond. Instead of a value of 2.26 Å as previously reported (Scatturin & Turco, 1958), a bond length of 2.15 Å corresponding to the sum of the covalent radii (Pauling, 1960) was selected. Consequently

Table 1. Assumed molecular dimensions

P---O	1.64 Å	OPO	101°
C---O	1.14	POP	124
Ni---C	1.82		
Ni---P	2.15		

Table 2.

	Wyckoff positions		Atomic coordinates (Å)		
			X	Y	Z
Ni	4 (e)	$X_1X_1X_1$	2.265	2.265	2.265
P	4 (e)	$X_2X_2X_2$	1.024	1.024	1.024
O(1)	6 (f)	X_300	1.794	0	0
O(2)	12 (i)	$X_4X_4Z_1$	1.672	1.672	5.104
C	12 (i)	$X_5X_5Z_2$	1.904	1.904	4.012

the values shown in Table 1 were adopted with linear Ni---C---O groups and tetrahedral valencies around the nickel atom. These parameters yield the coordinates shown in Table 2.

A calculation of the 19 available structure factors demonstrated that the model represented in Fig. 1 was correct. The scattering factors were taken from *International Tables for X-ray Crystallography* (1962) with the Ni curve corrected for the real part of the dispersion. Attempts were made to refine the model by least-squares but no satisfactory convergence could be obtained. This was due to the paucity of observed reflexions and the inability to measure more accurately the true intensities because of the relative low degree of crystallinity and possible effects induced by preferred orientation during sample preparation. All calculations were carried out on an Elliott 803B computer with the programming system of Daly, Stephens & Wheatley (1966).